## The Geometry of Coincidence-Site Lattices

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The density of coincidence sites in cubic crystal lattice planes in coincidence-site-related crystals with a multiplicity less than or equal to 31 has been investigated. A comprehensive table of lattice planes with a high density of coincidence sites for each of the twenty-two coincidence-site relationships is presented. In addition a complete list of the alternative ways of describing the twenty-two coincidence-site relationships as defined by Friedel in 1926 is presented.

#### 1. Introduction

There is now considerable evidence that the coincidence-site lattice defined by the common sites of two interpenetrating lattices is important in relation to the structure of grain boundaries in cubic crystals (Brandon, 1966). In particular, field-ion microscopy studies of the refractory metals (Brandon, Ralph, Ranganathan & Wald, 1964; Morgan & Ralph, 1967) have shown that the two crystals adjacent to a grain boundary often possess lattice sites defining a superlattice (the coincidence-site lattice) common to both crystal lattices. These experiments also indicate that such grain boundaries consist of a series of steps, and that the



Fig. 1. The  $(1\overline{10})$  plane of projection of the matrix and annealing twin lattices of a face-centred cubic crystal. The matrix, twin and coincidence-site lattice sites in the plane of the paper are represented by closed squares, circles and triangles respectively, and the matrix, twin and coincidence-site lattice sites in the next parallel (110) plane above and below the plane of the diagram are represented by open squares, circles and triangles respectively. The traces of the (111), (112) and (117) planes are indicated by T(111), T(112) and T(117) respectively. The directions [112], [111] and [772] are parallel to T(111), T(112) and T(117) respectively.

steps are planes that contain a high density of coincidence sites. The importance of the coincidence-site lattice concept is also evident from the results of recent electron-microscopy studies. Sargent (1968) has shown that steps in annealing twin boundaries in aluminum crystals define planes of the matrix and twin crystals for which all lattice sites are sites of the coincidence site lattice defined by the matrix twin crystal orientation relationship. Similar observations in austenitic stainless steel have been reported more recently by Vaughan (1969). Schober & Balluffi (1970) have made quantitative observations of orthogonal grids of grain boundary misfit screw dislocations in (001) twist boundaries in gold. These authors have shown that in the vicinity of coincidence site relationships the geometrical properties of the grids are consistent with those predicted for a boundary structure consisting of a suitable misfit dislocation grid embedded in the appropriate high density coincidence site interface.

As an aid to the interpretation of experimental results, there is a requirement in the case of cubic crystals for a comprehensive set of tables of coincidencesite relationships and a table of planes possessing a high density of coincidence sites for each of the possible coincidence site relationships. Both of these tables are presented in this paper and apply to primitive, body-centred and face-centred cubic crystals.

When a rotational symmetry operation is applied to a point lattice, a lattice is produced which is completely coincident with the original lattice. However, the application of a rotation other than a rotation associated with a symmetry element of the lattice may result in partial coincidence of the two lattices. The lattices related by such a rotation have certain lattice sites in common, which in themselves define a lattice termed the coincidence-site lattice.

Friedel (1926) has shown that two identical cubic lattices that are related by a coincidence site lattice may be considered to be related by a rotation of 180° about the normal  $[h_1h_2h_3]$  to a rational plane. The fraction of sites of each of the two interpenetrating lattices which are coincidence sites – that is sites which are common to both lattices – is equal to  $\Sigma^{-1}$ , where the multiplicity  $\Sigma$  is given by  $\Sigma = (h_1^2 + h_2^2 + h_3^2)$ , or  $\Sigma = (h_1^2 + h_2^2 + h_3^2)/2$  for  $(h_1^2 + h_2^2 + h_3^2)$  odd or even respectively.

Annealing twins in the face-centred cubic metals for example possess a coincidence-site lattice of multiplicity  $\Sigma = 3$ . The parent crystal and annealing twin lattices are related by a rotation of 180° about a  $\langle 111 \rangle$ direction, and the interface between the two crystals is usually the {111} plane normal to the axis of rotation. The coincidence-site lattice associated with the twin related crystals is illustrated in Fig. 1, which shows the (110) plane of projection plot of both parent and twin lattices. The (110) face-centred cubic planes have a two-fold stacking sequence, the parent, twin and coincidence site lattice points in the plane of the diagram are represented by closed squares, circles and triangles respectively, whereas the lattice points in the next parallel (110) plane above and below the plane of the diagram are represented by open squares, circles and triangles respectively. The trace of the interface plane between the twin and parent crystals is represented by a bold continuous line. It is to be noted that the interface plane may be chosen so that all lattice points in this plane are coincidence sites. Planes which consist only of coincidence sites are expected to have the lowest energy and therefore are the most likely to form the boundary between two crystals. Examples of planes in addition to the (111) plane in which all sites may be coincidence sites are the  $(11\overline{2})$  and (117) planes whose traces are represented by bold broken lines in Fig. 1. These planes have also been shown to be interface planes between parent and annealing twin crystals in aluminum (Sargent, 1968). The maximum fraction of sites in the lattice plane of particular Miller indices which can be coincidence sites will be represented by

 Table 1. Planes with a high density of coincidence sites for the twenty-two coincidence-site relationships with a multiplicity less than or equal to 31

The twenty-two coincidence-site relationships are described by a rotation of 180° about the direction with indices given in column 2 of the Table. A detailed description of the remainder of the Table is given in the text.

plicit	y rota	rotation		w <sup>1</sup> j													
3	111	(112)	101	Tu	στι	111 241 125 145 155 227	211 214 522 542 721 544	121 142 252 524 712 454	112 124 225 452 271, 445	221. 511. 441 425. 217.	2 <b>1</b> 2, 151, 414 254, 172,	122, 115, 144 245, 127,	411, 52I 541 711 552,	141, 512 514 171 525,	114. 251 451 117 255.	421 215 415 551 722	412 152 154 515 272
5	012	(031)	100	021	0ī2	012 ±543.	02ī • 534.	031 * 542	01 3 ± 524	043,	034.	07ī.	017,	± 512	±52ī	±531	\$51 <del>3</del>
7	123		120	11 <b>2</b>	<b>ī</b> 01	123 623,	3ī2 362,	231 236,	<b>4</b> 12	24Î	124	531	315	153	541	415	ī54
9	122	(411)	021	112	Ĩ <b>02</b>	1 <b>22</b> 112ь	411 221	511a 212	254 141.	245 114,	172ь 151	127c 115	255d	:3:	Ĩlla	211d	12īc
n	113	(332)	101	<b>ī</b> 31	0ī3	113	332	225	441								
13a	023	(051)	100	032	023	023	032	051	015								
13b	134		131	112	ī02	134	413	341									
15	125		121	212	<b>ī</b> 02	125 :5: 1513	245e Îllı 1153	:3: 2111	₹10 12Ĭ	120e 1121	3101 2213	1301 2123	315 122,	3151 4113	1351 1413	135 114 1	5113
17a	014	(053)	100	<b>04</b> 1	014	014	0 <b>4</b> ī	053	035								
17ь	223	(334)	102	<b>ī</b> 32	023	223	334	117	55 <b>ī</b>								
19	133	(611)	0 <u>3</u> 1	113	<b>ĩ</b> 03	133	<del>6</del> 11	722	544								
2la	145		110	144	131	145 2111	5ī4 ī21 <sub>1</sub>	451 1121	:3: 2213	321 2123	213 1223	I32 4113	5131 1413	351 1 1143	1351 5113	:7: 1513	11Ĩ 1153
2Њ	124		0 <b>2</b> 1	212	I04	124 1121	:3: 2213	3121 2123	231 1223	1231 4113	5311 1413	315 1143	153 5113	:7: 1513	111 1 115 1	211	121 1
23	136		031	212	<u>ī</u> 03	136											
25a	034	(071)	100	043	034	034	043	07 1	017	:5:	0211	01 <b>2</b> 1	0311	013 †	± 531 1	± 513 1	
25b	345		251	102	234	345	:5:	2101	<b>120</b> 1	3101	1301	3151	<b>3</b> 15 t	1351	1351		
27a	127		274	<b>T</b> 01	0ī3	127 1 <b>14</b> ,	:3: 151	1221 115	4111	51 <b>T</b>	:9:	2111	1211	ī12 t	2213	2123	141
27ь	115	(552)	111	141	015	115 141	552 5113	:3: 1513	2211	1141	:9:	2111	ī21 <sub>1</sub>	1121	2123	1223	<b>4</b> 1Ī
29a	234		132	223	104	234											
29b	027	(073)	100	052	025	025	05 <b>2</b>	073	037								
3la	156		152	113	<b>ī</b> 03	156	561	615									
31ь	237		320	<b>2</b> Ī1	014	237											

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 $\sigma^{-1}$ , where  $\sigma$  is a rational factor of  $\Sigma$  being equal to one when all sites in a lattice plane are coincidence sites. In § 2 of this paper a procedure for determining the value of  $\sigma$  for a particular lattice plane is described, and the planes with  $\sigma$  values less than  $\sigma = \Sigma$  are tabulated for a wide range of Miller indices for all coincidence site relationships with  $\Sigma \leq 31$ .

The orientation relationship between the parent and twin crystals illustrated in Fig. 1 may, for example, also be described by rotation of 180° about [112] or 70.5° about [110]. In general there are twenty-four different ways of describing the orientation relationship between two cubic crystals and hence twenty-four ways of describing the orientation relationship between two coincidence-site-related crystals. In § 3 of this paper the different ways of describing the orientation relationship between the coincidence-site-related crystals are tabulated for the coincidence-site relationship with  $\Sigma \leq 31$ . The ways in which the table may be used to aid the interpretation of experimental results is also described.

# 2. The density of coincidence sites in a crystal lattice plane

All coincidence-site lattices with a multiplicity  $\Sigma$  less than or equal to 31 have been considered, and the coincidence-site lattice relationships which arise in the range  $\Sigma \leq 31$  are listed in Table 1 where  $\Sigma$  is given in column 1 and the corresponding axis of rotation  $[h_1 h_2 h_3]$  is given in column 2. For some values of  $\Sigma$ two coincidence-site relationships which are not crystallographically equivalent arise; they are represented by  $\Sigma_a$  and  $\Sigma_b$ . In some cases a rotation of 180° about axes of a different form give rise to crystallographically equivalent relationships, as for example when the two axes are orthogonal and together define a symmetry plane. These alternative descriptions are given in column 3 of Table 1.

The following procedure was used for determining the Miller indices of planes parallel to a plane in the crystal lattice in which all sites can be sites of the coincidence-site lattice, or in which  $\sigma^{-1}$  of the crystal

# Table 2. The alternative ways of describing the twenty-two coincidence-site relationships with a multiplicity less than or equal to 31

An asterisk indicates the description associated with the smallest angle of rotation for a particular multiplicity.

$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	Axis of rota- tion	Multipli- city	Angle of rota- tion	•														
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	100	5 *	36.9	310	5	180	411	9	180 511	7	158.2 610	19a	161.3 551	13a	164.1	733	17b	166.1
$\begin{array}{c c c c c c c c c c c c c c c c c c c $		13a*	22.6		7	115.4		11	129.5	9	120.0	23	127.5	15	134.4		19 <b>b</b>	139.7
$\begin{array}{cccccccccccccccccccccccccccccccccccc$		17a*	28.1		11	144.9		17a	93.4	13a	92.2	3la	101.2	19b	110.0		23	117.2
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$		25a*	16.3		13b	76.7		19b	153.5	19a	73.2		100.0	250	91.2		290	98.9
$ \begin{array}{c c c c c c c c c c c c c c c c c c c $		<b>29</b> b*	43.6		19a	93.Q		27a	109.5	27a	60.0 532	190	180.0	101.				
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	110				23	55.6		27b	70.5	275	180.0	215	144.1 711	130	164.1	751	192	166.8
$\begin{array}{c c c c c c c c c c c c c c c c c c c $		3	70.5					_		316	137.9	27a	114.0	10-	134.4		210	141.8
$\begin{array}{c c c c c c c c c c c c c c c c c c c $		9.*	38.9	311	3	146.4	331	5	154.2		150 0 011	10.	100.0	192	110.0		250	120.0
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$		11 *	50.5		-5	95.7		7	110.9 432	15	159.0 611	192	180.0	25a	91.2		310	102.1
$\begin{array}{c c c c c c c c c c c c c c c c c c c $		17b	86.6		9	67.1		11	82.2	19a	121.8	210	144.1	071				
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$		19a*	26.5		11	180		17b	63.8	27a	94.3	270	114.0 641	270	164.4	753	210	167.5
$\begin{array}{c c c c c c c c c c c c c c c c c c c $		27b*	31.6		15	50.7		19a	180	29a	180.0			310	135.2		23	143.6
$\begin{array}{c c c c c c c c c c c c c c c c c c c $					15	117.8		23	130.7		443	215	162.3	07			275	122.5
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$	111	3 *	60		23*	40.5		25b*	51.7 520	15	159.0	25a	129.8 720	272	164.4			
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$		7 *	38.2		25b	168.3				196	121.8			310	135.2	911	21a	167.5
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$		13b*	27.8		27a	79.3	421	11	155.4	27b	94.3 540	21a	162.3				23	143.6
$\begin{array}{cccccccccccccccccccccccccccccccccccc$		19b*	46.8		31b	126.6		15	113.6	29b	180.0	255	129.8 552	275	180.0		27a	122.5
$\begin{array}{cccccccccccccccccccccccccccccccccccc$		2la*	21.8					21b	180			- 11		296	149.6			
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$		3la*	17.9	320	7	149.0		23	85.0 521	15	180.0 621	210	162.3			931	23	168.0
$\begin{array}{cccccccccccccccccccccccccccccccccccc$					11	100.5		25b	132.8	175	139.9	25b	129.8 721	27a	180.0		25b	145.1
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	210	3	131.8		13a	180				23	107.7			29a	149.6		29a	124.7
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$		5	180		17b	122.0	332	11	180	31b	159,3 541	2la	180.0					
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$		7	73.4		19b	71.6		13a	133.8			23	145.7 544	<b>2</b> 9a	164.9	755	25b	168.5
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$		9	96.4		29a	84.1		19a	99.1 441	17a	160.3	29a	116.6				27b	146.4
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$		15*	48.2		31b	54.5		23	155.9	2lb	124.9		722	29a	164.9		3la	126.6
$\begin{array}{cccccccccccccccccccccccccccccccccccc$		21b	58.4					29a	76.0	29a	97.9 533	11	162.7					
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$		23	163.0	321	7	180		31a	114.8			13b	130.8 730	29b	180.0	771	25a	168.5
$\begin{array}{c c c c c c c c c c c c c c c c c c c $		27a*	35.4		9	123.8			522	17b	160.3	17a	105.3	31b	150.6		27a	146.4
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$		29a	112.3		15	86.2	430	136	157.4	21b	124.9	23	86.3				31ь	126.6
$\begin{array}{c ccccccccccccccccccccccccccccccccccc$					15	150.1		17b	118.1	29b	97.9	31b	72.2 553	15	165.2			
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$	211	3	180		23	102.6		25a	180					17a	137.3	773	27a	169.0
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$		5	101.5		25b	63.9		25b	90 433	17b	180.0 542	23	163.0	2la	113.9		29b	147.7
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$		7	135.6					29a	136.4	19a	142.1	27a	131.8	27b	95.3			
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$		11	63.0	322	9	152.7				25a	111.1					951	27a	169.0
$ \begin{array}{c ccccccccccccccccccccccccccccccccccc$		15	78.5		13a	107.9	431	13b	180		631	23	180 731	15	165.2		29a	147.7
$ \begin{array}{cccccccccccccccccccccccccccccccccccc$		<b>2</b> lb*	44.4		17b	180		15	137.2 530	17a	180.0	25b	147.1	17b	137.3			
$\begin{array}{cccccccccccccccccccccccccccccccccccc$		25b	156.9		2la	128.3		21b	103.8	19b	142.1	31b	118.9	21b	113.9	95 <b>3</b>	29a	169.4
3lb*         52.2         3lb         80.7         632         25b         163.7           410         9         152.7         531         9         160.8         29a         133.6         645         31b         165.4         775         31b         169.7           221         5         143.1         13b         107.9         510         13a         180         11         126.2           9         90         17a         180         15         137.2         15         99.6         543         25b         180.0         650         31a         165.4         11.1         31a         169.7		29a	149.6		2lb	79.0		27a	157.8	25b	111.1			27a	95.3		31b	148.7
410 9 152.7 531 9 160.8 29a 133.6 645 31b 165.4 775 31b 169.7 221 5 143.1 13b 107.9 510 13a 180 11 126.2 9 90 17a 180 15 137.2 15 99.6 543 25b 180.0 650 31a 165.4 11.1.1 31a 169.7		31b*	52.2					31b	80.7		632	25b	163.7					
221 5 143.1 13b 107.9 510 13a 180 11 126.2 9 90 17a 180 15 137.2 15 99.6 543 25b 180.0 650 31a 165.4 11.1.1 31a 169.7				410	9	152.7			531	9	160.8	29a	133.6 645	31ь	165.4	775	31b	169.7
9 90 17a 180 15 137.2 15 99.6 543 25b 180.0 650 31a 165.4 11.1.1 31a 169.7	221	5	143.1		13b	107.9	510	13a	180	11	126.2							
		9	90		17a	180		15	137.2	15	99.6 543	25b	180.0 650	3la	165.4	11, 1, 1	3la	169.7
9 180 21a 79.0 21a 103.8 21b 80.4 27a 148.4		9	180		2la	79.0		2la	103.8	2lb	80.4	27a	148.4					
13b 112.6 21b 128.3 27b 157.8 29a 66.6 732 31b 180.0		13b	112.6		21b	128.3		27b	157.8	29a	66.6		732	31b	180.0			
17b* 61.9 31a 80.7 710 25a 180.0		17b*	61.9					3la	80.7		710	25a	180.0					
2bb 73,7 27a 148.4 651 31a 180,0		25b	73.7									27a	148.4 651	3la	180.0			

177

lattice sites can be coincidence sites for  $\sigma$  a rational factor of  $\Sigma$  not equal to  $\Sigma$ . For a particular coincidence site relationship the value of  $\sigma$  is the same for a particular plane in all three cubic Bravais lattices. For convenience the primitive cubic lattice will be considered.

The Miller indices of planes and directions in one of the crystal lattices (lattice 1) of the coincidencesite-related lattices will be referred to the orthonormal cubic lattice basis  $c_1$ ,  $c_2$ ,  $c_3$ . For convenience the two basis vectors  $\mathbf{d}_1$  and  $\mathbf{d}_2$  of the basis  $\mathbf{d}_1$ ,  $\mathbf{d}_2$ ,  $\mathbf{d}_3$  defining a primitive cell of the coincidence lattice are chosen to be in the plane with normal  $[h_1, h_2, h_3]$ , and hence in cubic crystals the plane with Miller indices  $(h_1 h_2 h_3)$ . This plane is a lattice plane of the two crystal lattices and of the coincidence-site lattice. Procedures for determining the two vectors  $\mathbf{d}_1 = l_i \mathbf{c}_i$  and  $\mathbf{d}_2 = m_i \mathbf{c}_i$  which must define a primitive cell in the  $(h_1 h_2 h_3)$  plane have been described by Jaswon (1965) and Bevis (1969). When the lattice planes  $(h_1 h_2 h_3)$  have an odd stacking sequence (Jaswon & Dove, 1955) the vector  $\mathbf{d}_3$  is chosen to be normal to this plane, so that  $\mathbf{d}_3 = h_i \mathbf{c}_i$ . In the case of a plane with an even stacking sequence the vector  $\mathbf{d}_3$  may be taken to be any primitive vector of the vectors  $\frac{1}{2}(l_i+m_i+h_i)\mathbf{c}_i$ ,  $\frac{1}{2}(l_i+m_i)\mathbf{c}_i$ ,  $\frac{1}{2}(l_i+h_i)\mathbf{c}_i$  or  $\frac{1}{2}(m_i+h_i)\mathbf{c}_i$ , and in general will be represented by  $\mathbf{d}_3 = n_i \mathbf{c}_i$ . The relationship between components of a vector referred to the orthonormal crystal lattice basis and the coincidence site lattice basis, represented by  $\mathbf{x} = u_i \mathbf{c}_i$  and  $\mathbf{x} = v_i \mathbf{d}_i$  respectively is given by  $v_i = w_{ij} u_j$ where

$$w_{ij} = \begin{pmatrix} l_1 \ m_1 \ n_1 \\ l_2 \ m_2 \ n_2 \\ l_3 \ m_3 \ n_3 \end{pmatrix}.$$

The corresponding transformation of Miller indices is given by

$$b_i = w_{ji}a_i , \qquad (1)$$

where  $(a_i)$  and  $(b_i)$  are the Miller indices of a plane referred to the crystal lattice and coincidence site lattice bases respectively. If the vector  $b_i \mathbf{d}_i$  is a primitive lattice vector, then the interplanar spacing of the planes  $(b_1 b_2 b_3)$  is equal to the interplanar spacing of the parallel planes  $(a_1 a_2 a_3)$ , so that only  $1/\Sigma$  sites in the plane  $(a_1 a_2 a_3)$  are coincidence sites. If however  $b_i \mathbf{d}_i$ is not a primitive reciprocal lattice vector, and is equal to  $b_i \mathbf{d}_i = pb'_i \mathbf{d}_i$  where p is an integer and  $b'_i \mathbf{d}_i$  is a primitive reciprocal lattice vector, then the interplanar spacing of the planes  $(a_1 a_2 a_3)$  is then 1/p that of the planes  $(b'_1 b'_2 b'_3)$ . In this case a fraction  $p/\Sigma = 1/\sigma$  sites in the plane  $(a_1 a_2 a_3)$  are coincidence sites. When  $p = \Sigma$  then all sites are coincidence sites.

The densities of coincidence sites in parallel lattice 1 and lattice 2 planes are in general not equal. The density of coincidence sites in a lattice 2 plane can be determined in exactly the same way as for lattice 1 planes. The Miller indices of the plane in lattice 2 parallel to the  $(a_1 a_2 a_3)$  plane in lattice 1 has the Miller

indices  $(e_1 e_2 e_3)$  with respect to the lattice 2 basis given by  $e^i = r_{ji}a_j$  where  $r_{ij} = -\delta_{ij} + 2h_ih_j(h_kh_k)^{-1}$ . The same matrix  $w_{ij}$  applies for lattice 2 and lattice 1.

With the aid of a digital computer the authors have used equation (1) to determine the density of coincidence sites in crystal lattice planes for the coincidence site relationships given in column 2 of Table 1. The three rows of the matrices  $w_{ij}$  for the twenty-two coincidence site relationships are given in columns 4, 5 and 6 of Table 1. For each coincidence-site relationship the Miller indices of the following crystal lattice planes are given:

(a) planes  $(a_1 a_2 a_3)$  in which all sites can be coincidence sites, but with the limitation that only the fifty-four most densely packed primitive cubic lattice planes are considered, (b) planes  $(a_1 a_2 a_3)$  in which a fraction  $1/\sigma$ sites can be coincidence sites, but with the limitation that only the first nine most closely packed primitive, body-centred and face-centred cubic lattice planes are considered. The variants of planes given in Table 1 apply for the description of the coincidence-site relationship given in column 2 of the Table. The planes are given in columns 7-18 of Table 1 and are listed across the page, the planes with a fraction  $1/\sigma$  of coincidence sites are preceded by  $:\sigma$ : A difference in the density of coincidence sites in parallel lattice 1 and lattice 2 planes is indicated in the following way. If the density of coincidence sites in the lattice 2 plane parallel to the lattice 1 plane given in Table 1 is equal to  $1/\Sigma$ , then the indices of the lattice 1 plane are followed by a dot. If the Miller indices of a lattice 1 plane are followed by a number, then the reciprocal of the number is the density of coincidence sites in the parallel lattice 2 plane. In general the indices of the parallel lattice 2 planes do not fall within the range of planes investigated. However in cases where both planes fall within the range investigated the pairs of planes are followed by the same letter.

# 3. Alternative descriptions of coincidence-site relationships

In an experimental determination of the orientation relationship between two crystals of interest, for example in the case of two coincidence-site related crystals. the orientation relationship could be that described by column 2 of Table 1 or in twenty-three other ways which describe the same orientation relationship (Lange, 1967), and which result when any particular original rotation is combined with the twenty-four proper symmetry rotations associated with a cube having identical faces. Thus, unless the twenty-four ways of describing the orientation relationship determined experimentally are deduced, and then compared with the relationships of column 2 of Table 1, it is not possible to establish readily that a coincidence site relationship is operative. Only a limited number of the alternative ways of describing the orientation relationships of column 2 of Table 1 have been given previously. The

most comprehensive list, which is, however, limited in extent, has been given by Ranganathan (1967), and are the results of an application of a procedure (Ranganathan, 1966) for determining the possible  $\Sigma$ values and corresponding angles of rotation associated with a particular axis of rotation. The authors have determined the axes of rotation and the corresponding angles of rotation ( $\omega$ ) for all twenty-four ways of describing each of the twenty-two coincidence-site relationships given in column 2 of Table 1. These results are presented in Table 2 with the limitation that only the form of the axes of rotation are given. However, an experimentally determined orientation relationship may be compared with the relationships presented in Table 2, thus allowing the operative coincidence-site relationship to be determined readily. The more convenient orientation relationship given in column 2 of Table 1 may then be adopted, and the further crystallographic information contained in Table 1 utilized directly.

In Table 2 the results are presented in multiples of three colums, the first, second and third column in every three gives the axis of rotation, the multiplicity and the angle of rotation ( $\omega$ ) respectively. For the [100], [110] and [111] symmetry axes the angles of rotation (90- $\omega$ , 90+ $\omega$ , 180- $\omega$ ), (180- $\omega$ ) and (120  $-\omega$ , 120+ $\omega$ ) respectively also give rise to the coincidence-site relationships indicated in column 2. The relationship which gives the smallest value of  $\omega$  for a

particular value of  $\Sigma$  is indicated by an asterisk. Table 2 effectively gives the results of an application of the procedure described by Ranganathan (1966) for all possible rotations about axes with indices  $[l_1 \ l_2 \ l_3]$  for  $\{(l_1)^2 + (l_2)^2 + (l_3)^2\} < 123$  which give rise to the coincidence site relationships with  $\Sigma \leq 31$ .

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### The Effect of Refraction in the Small-Angle Diffraction of X-rays from Stacked Lamellae

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An expression for the scattering of X-rays from a regular stack of lamellae is developed, releasing the prior restraint that no refraction may occur. The effect of refraction is to cause deviations from the classical Bragg condition. Further, a condition of total reflection is shown to occur within systems whose lamellar spacings are greater than a critical value.

#### Statement of the problem -

The analysis of X-ray scattering data generally proceeds through a consideration of the amplitude of scattering A(S) as given by

$$A(\mathbf{S}) = A_e(\mathbf{S}) \int \varrho(\mathbf{X}) \exp \{2\pi i \mathbf{S} \cdot \mathbf{X}\} dv_{\mathbf{x}} .$$
(1)

Here,  $A_{\ell}(S)$ ,  $\rho$ , X, S, and  $v_x$  have their usual meanings: amplitude of scattering by an isolated electron, electron

A C 27A – 6\*

density, real space vector, reciprocal space vector, and real space volume. This expression is rigorously correct (in the kinematic approximation) only if (a) the medium is non-absorbing and (b) the vector S is permitted to vary as the X-ray beam traverses regions of differing refractive index. Indeed, absorptive and refractive corrections are not needed in the analysis of diffraction line breadths and peak positions; *e.g.*, the relative errors in atomic positional analysis are of the order of  $10^{-5}$  and can safely be ignored. However, for the particular case of diffraction from a lamellar system whose elements are 100 Å or more in thickness both absorptive and refractive effects can become large. The under-

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